A Comparative Study of V₆O_{13.27} and V₆O₁₃

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A comparative study of the oxygen-enriched $V_6O_{13,27}$ and stoichiometric V_6O_{13} is made from susceptibility and ⁵¹V NMR. Instead of a susceptibility peak at 150 K as in V_6O_{13} , a plateau was observed in this region. A single broad ⁵¹V NMR line was observed in $V_6O_{13,27}$ at room temperature, which narrowed on lowering of temperature. Close to 150 K, three different lines, corresponding to the three inequivalent sites as in V_6O_{13} , were observed. Though occurrence of spin-pairing at 150 K has been recognized for some time, a bipolaronic mechanism for this pairing in V_6O_{13} itself is proposed for the first time. Interestingly, though susceptibilitywise $V_6O_{13,27}$ and V_6O_{13} show very different behavior, bipolaronic spin-pairing possibly occurs in $V_6O_{13,27}$ also. However, at 50 K, antiferromagnetic ordering as in $V_6O_{13,27}$ as the latter has fewer V⁴⁺ spins than required, due to the presence of excess oxygen. © 1989 Academic Press, Inc.

Introduction

In recent years, V_6O_{13} has been found to be one of the best cathode materials in conjunction with polymeric electrolytes, suitable for use in lithium-based solid-state batteries (1). Stoichiometric V_6O_{13} has the formal oxidation assignment of $V_4^{4+}V_2^{5+}$ but behaves as a mixed-valent compound (2, 3). V_6O_{13} at room temperature has a monoclinic structure (Fig. 1) with space group C/2m. Thus, one has nominal $V^{5+}O_6$ octahedra (V₂) sandwiched between nominal $V^{4+}O_6$ octahedra (4) involving V(1) and V(3).

V₆O₁₃ exhibits metallic luster with conductivity 10^{-1} (ohm cm)⁻¹ and has been described in the literature as a metal (2, 5). The sharp drop in χ at 150 K has been ascribed to a metal insulator (M-I) transition, with a crystallographic distortion (4) from a symmetry of C/2m to C/2. The lattice parameters also show the corresponding discontinuous changes (4). The shortest V-Vdistance in V_6O_{13} is for V(2)-V(3) along the double zigzag ribbon (3.009 Å); next, one encounters the V(1)-V(1) separation across shared octahedral edge (3.049 Å). The largest separation is for V(2)–V(3) in the a-cplane (3.38 Å). The temperature dependence of conductivity (σ) of pure V₆O₁₃ (Fig. 2) is that of a semiconductor (6) and not that of a metal. Thus from room temperature down to 150 K, there is a sudden drop near 150 K, then again a semiconductor-like behavior down to 90 K.

The present paper deals with a comparative study of stoichiometric V_6O_{13} and

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FIG. 1. Crystal structure of V₆O₁₃.

 $V_6O_{13,27}$, particularly to see how excess oxygen atoms modify the V_6O_{13} lattice and its properties. These systems have been investigated by both a macroprobe magnetic susceptibility and a microprobe ⁵¹V NMR.

Experimental

Nonstoichiometric V₆O_{13.27} was prepared by the decomposition of NH₄VO₃ in a longnecked glass ampoule (7). The extent of nonstoichiometry of the oxide was measured by measuring the weight gain of the specimen during its oxidation in air in a thermal analyzer (Stanton Redcroft, STA 780 UK) and the exact formula was found to be $V_6O_{13,27}$ (8). Susceptibility measurements were carried out with a Squid magnetometer over the temperature range 2-400 K and in fields of 0.5-5 T. The NMR spectra were taken at 15 MHz on a Varian WL 210 NMR spectrometer with a Varian lowtemperature accessory. The ⁵¹V shifts reported are with reference to Na₃VO₄ in solution.

Results and Discussion

Figure 3 sums up the susceptibility results of the system. The sharp break in χ in pure V_6O_{13} at 150 K is considerably rounded off even in $V_6O_{13,05}$ (9) and vanishes altogether in $V_6O_{13,27}$; a plateau has been observed near this temperature. There is a steady increase in χ below 150 K, where it follows the Curie-Weiss law. The results in the case of $V_6O_{13,27}$ are similar to those observed by Jhans and Honig (10) in the case of nonstoichiometric cation deficient V_2O_3 ($V_{2-\nu}O_3$). Here the cation deficiency causes the lattice to distort (contract), counteracting the high-temperature lattice expansion. As such, one can infer that an excess of oxygen in the case of $V_6O_{13,27}$ possibly causes the lattice to distort and hence flattening is seen in the $\chi - T$ plot.

⁵¹V NMR

⁵¹V NMR of polycrystalline V₆O₁₃ at 300 K (Fig. 4) simulates the V₆O₁₃ single-crystal spectrum (5) but obviously is not as sharp and not all satellites corresponding to $I = \frac{\tau}{2}$



FIG. 2. Electrical conductivity vs temperature for oxides of vanadium.





are clearly resolved. But the spectrum does clearly reveal three nonequivalent sites in the shifts, viz. V_1 (-1.4%) at highest field,

 V_3 (-1%) in the middle, and V_2 (-0.09%) at the extreme left. These shifts parallel the charge distribution of 4.16, 4.34, and 4.6 for



FIG. 4. 51 V spectra at 15 MHz. (a) V₆O₁₃ with signal averager at 300 K; (b) V₆O₁₃ with single scan; (c) V₆O_{13.27}.

 V_1 , V_3 , and V_2 , respectively, obtained from X-ray data (4). Across the phase transition at 150 K, the charge distribution changes to 4.4, 4.4, and 4.2 for V₁, V₂, and V₃, respectively (4). The NMR shifts, viz. V_1 (0.1%) and V_3 (-1.3%), corroborate this changed charge distribution. Thus, one observes an intense low-field line from overlapping V_1 and V_2 and a high-field wide line from V_3 (which has become most V⁴⁺-like). NMR shifts and X-ray data indicate that V₁ spins are preferentially paired in V_6O_{13} at 150 K, whereby the charge distribution changes drastically from 4.16 to 4.4. Thus it is the V₃ spins which become antiferromagnetic at 50 K.

In the case of $V_6O_{13,27}$ no satellites are present in ⁵¹V NMR at 300 K and a single broad asymmetric line is obtained, indicating a distortion of the V_6O_{13} lattice itself. But X-ray values for V₆O_{13.27} agreed well (8) with the values for V_6O_{13} reported in the literature (4). However, NMR as a more sensitive probe is able to detect subtle distortions, particularly in the quadrupolar interaction. On lowering the temperature, the ⁵¹V line increasingly narrows and becomes more intense and at 143 K an almost symmetric intense narrow central line along with two broad (weak) lines appear on either side. Frequency variation study from 15 to 8 MHz shows the line positions to be frequency-dependent. However, the narrowing of the broad line at 300 K, with lowering of temperature, indicates spin-pairing. Though a structural transition at 150 K as in V₆O₁₃ may not occur, narrowing leads to reemergence of the three lines corresponding to three vanadium sites V_1 , V_2 , and V_3 , broadened by the distortion due to extra oxygen at 300 K.

Conclusion

From the χ -T diagram (Fig. 3) it is evident that the spin-paired state at 150 K causing the so-called *M*-I transition is

present in V_6O_{13} . This is not disturbed even by the antiferromagnetic transition at 50 K. But so far no mechanism for such spin-pairing has been suggested. Goodenough (11) already established a critical overlap criterion for predicting localized vs collective electron behavior. In 3d metal oxides, the critical separation R_c (max) is 2.97 Å and as such spin-pairing results in M-I transition of VO₂, where the V-V distance is 2.85 Å. As pointed out before, the V(1)-V(1) separation in V_6O_{13} is 3.049 Å, which is significantly larger than $R_{\rm c}$. So, considering the V-V distances, the susceptibility peak in χ -T diagram, and intensification of NMR lines with narrowing, it is suggested here that this strong spin-paired state at 150 K arises from the same type of "bipolaronic mechanism" (12-17) that has been postulated in β -Na_{0.4}V₂O₅, β -Li_{0.4}V₂O₅, and $Pb_rV_2O_5$ (18). Moreover, the X-ray data (4) of V_6O_{13} , which indicates that V_1 is displaced away from the mirror plane at $T_{\rm c}$, may be cited as an indication of phononmediated distortion in the bipolaronic state. It thus appears that though the structural distortion in the oxygen enriched $V_6O_{13,27}$ flattens out χ -T peak, the non-Curie behavior of χ , and also the appearance of three lines near 150 K, points to the same type of bipolaronic spin-pairing mechanism as in V_6O_{13} .

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